



# Hierarchical bicomponent TiO<sub>2</sub> hollow spheres as a new high-capacity anode material for lithium-ion batteries

Ruiping Liu<sup>1,2,3,\*</sup> , Chao Shen<sup>1</sup> , Chao Zhang<sup>1</sup> , James Iocozzia<sup>3</sup> , Qi Wang<sup>1</sup> , Shiqiang Zhao<sup>3</sup> , Kunjie Yuan<sup>3</sup> , and Zhiqun Lin<sup>3,\*</sup>

<sup>1</sup>Department of Materials Science and Engineering, China University of Mining and Technology (Beijing), Beijing 100083, China

<sup>2</sup>State Key Laboratory of Coal Resources and Safe Mining, China University of Mining and Technology (Beijing), Beijing 100083, China

<sup>3</sup>School of Materials Science and Engineering, Georgia Institute of Technology, Atlanta, GA 30332, USA

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## ABSTRACT

Hierarchical TiO<sub>2</sub>-based hollow spheres were successfully synthesized via a hydrothermal method using FeSO<sub>4</sub>·7H<sub>2</sub>O, CoSO<sub>4</sub>·7H<sub>2</sub>O and ZnSO<sub>4</sub>·7H<sub>2</sub>O as soft templates. The as-prepared hollow spheres are well dispersed with the diameters of 2–4 μm. The shell and the interior surface of the spheres are composed of loosely packed grains, which provide a large specific surface area to facilitate lithium-ion diffusion processes. Among the three types of hybrid hollow spheres, TiO<sub>2</sub>/Fe<sub>2</sub>O<sub>3</sub> shows the highest reversible capacity and best cycling stability (discharge capacities of 290.8 and 210.5 mAh/g were achieved after 100 cycles at 0.1C and 1C, respectively) and rate performance (from 461.1 mAh/g at 0.1C to 79.3 mAh/g at 5C with recovery to 288.6 mAh/g at 0.1C) for anode materials in lithium-ion batteries.

## Introduction

Ever increasing demands for energy have led to staggering rates of fossil fuel consumption and corresponding high rates of harmful emissions. With established and mounting concerns over climate change, efforts have been dedicated to developing new energy storage devices for renewable and clean energy technologies. As one of the most promising energy storage technologies, lithium-ion batteries (LIBs) are widely used as a convenient power source

for various portable electronics and electric vehicles (EV) due to their superior properties including relatively high-power density, long cycling life and environmental friendliness [1–4]. However, the limited energy density of lithium-ion batteries restricts their wider application in a variety of emerging vehicles. The energy density of a battery depends heavily on the output voltage and electrode theoretical capacity [5, 6]. It is well known that improvements in cathode performance are far more challenging when compared to anodes.

Address correspondence to E-mail: bjlrp165@126.com; 201402@cumt.edu.cn; zhiqun.lin@mse.gatech.edu

Consequently, anode development is one of the most important and accessible factors for improving LIB energy density and safety [7–9].

Compared to the commercial graphite-based anode materials with relative low theoretical capacity (372 mAh/g) and safety concerns, metal oxides such as  $\text{TiO}_2$  [10],  $\text{Fe}_3\text{O}_4$  [11],  $\text{Co}_3\text{O}_4$  [12],  $\text{SnO}_2$  [13, 14],  $\text{NiO}$  [15] and  $\text{MnO}$  [16] are promising candidates for the use in lithium-ion batteries due to their relatively high theoretical capacity, stable reversible capacity and safety. However, many metal oxides suffer from the large volume changes ( $> 300\%$ ) during charging and discharging leading to pulverization and delamination of the anode ultimately leading to poor cycling performance [17, 18]. In addition, the poor electrical conductivity of the metal oxides typically leads to electrodes with poor rate capacity [19, 20]. Among the various metal oxides,  $\text{TiO}_2$  has received particular interest as a promising battery anode candidate because of its low capacity loss, high-power capability, low cost, abundance and environmental friendliness [21]. Furthermore, the relatively small volume change ( $< 4\%$ ) during charging and discharging helps to preserve the electrode structure enabling prolonged cycling life. Although the working potential of  $\text{TiO}_2$  is a little high to be used as an anode material, it remains an attractive option due to its electrochemical stability in common organic electrolytes which leads to an improved overcharge protection and overall improved battery safety [22].

The electrochemical properties of  $\text{TiO}_2$  electrodes heavily depend on their composition, microstructure, crystallinity and crystallite size. It is well known that nanostructured materials possess enhanced cycling and rate performance. Nanostructured  $\text{TiO}_2$  with different morphologies, such as nanoparticles [23, 24], nanorods [25], nanotubes [26], nanowires [27], nanosheets [28] and nanospheres [29, 30], has been successfully synthesized, and the electrochemical properties extensively investigated. However, the poor electronic conductivity ( $10^{-12}$ – $10^{-7}$  S/cm), the frequent aggregation of  $\text{TiO}_2$  nanoparticles and the low  $\text{Li}^+$  diffusion rate ( $10^{-15}$ – $10^{-9}$   $\text{cm}^2/\text{s}$ ) often hinder the electrochemical performance of  $\text{TiO}_2$ -based electrodes. In order to improve the electric conductivity, carbonaceous materials such as amorphous carbon [31], carbon nanotubes [32] and graphene [33] are often used to form either a mixture with  $\text{TiO}_2$  or a thin coating layer on its surface. Despite the advantages of adding carbon, its introduction into  $\text{TiO}_2$  can

reduce the amount of active materials in the electrode resulting in a reduced mass energy density and volume energy density of resulting batteries. Recently, many works focused on fabricating  $\text{TiO}_2$ -based composite as anode materials for LIBs in order to improve the electrochemical performance of  $\text{TiO}_2$  by utilizing the synergistic effect between  $\text{TiO}_2$  and other metal oxides, such as  $\text{Fe}_2\text{O}_3$  [34],  $\text{Nb}_2\text{O}_5$  [35],  $\text{CoO}$  [36],  $\text{SnO}_2$  [37],  $\text{NiCo}_2\text{O}_4$  [38]. Among them, due to the high theoretical capacity of  $\text{Fe}_2\text{O}_3$  (1005 mAh/g),  $\text{Co}_2\text{O}_3$  (890 mAh/g) and  $\text{ZnO}$  (981 mAh/g), they show great prospect toward high-energy anodes, and it can be deduced that the capacity will be largely improved by preparing  $\text{TiO}_2$ -based composites with the above three metal oxides due to their synergistic effect. Moreover, due to their conversion reaction mechanisms during cycling, the metal oxide will be reduced to metal first, which will improve the electrical conductivity of the  $\text{TiO}_2$ -based anode materials, and finally enhance the electrochemical performance.

Herein, we present bicomponent porous hierarchical  $\text{TiO}_2$  spheres for facilitating lithium-ion diffusion through active materials by enlarging the contact area between the liquid electrolyte and the active materials. In addition, a second phase of metal oxide was introduced to the  $\text{TiO}_2$  grains by a one-step hydrothermal method using hydrated metal sulfates as soft templates. The  $\text{ZnO}$ ,  $\text{Fe}_2\text{O}_3$  and  $\text{Co}_2\text{O}_3$  will react with lithium ions to form the corresponding Zn, Fe and Co and  $\text{Li}_2\text{O}$  during charging first, which may facilitate electron and ion transfer during cycling and thus may also improve the electrochemical performance of the  $\text{TiO}_2$  as anode materials in lithium-ion batteries.

## Experimental

### Synthesis of $\text{TiO}_2$ hybrid hollow spheres

In a typical synthesis procedure, 1 g of  $\text{FeSO}_4 \cdot 7\text{H}_2\text{O}$ ,  $\text{CoSO}_4 \cdot 7\text{H}_2\text{O}$  or  $\text{ZnSO}_4 \cdot 7\text{H}_2\text{O}$  was added to anhydrous ethanol (60 ml) under continuous stirring for 1 h to form a white suspension. Tetrabutyl titanate (TBT) (4 ml) was then added to the suspension over 20 min. The mixture was stirred for 1 h and transferred to a 100-mL Teflon-sealed autoclave maintained at 200 °C for 24 h. The resulting precipitates were separated and purified by centrifugation and washed three times with DI water and ethanol.

Samples were then dried in a vacuum oven at 70 °C for 5 h to yield hierarchical porous  $\text{TiO}_2/\text{Fe}_2\text{O}_3$ ,  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$  hybrid hollow spheres.

### Characterization

X-ray diffraction (XRD) patterns were measured on a Bruker D8 Advance diffractometer using Ni-filtered  $\text{Cu K}\alpha$  radiation. The morphologies of the as-prepared bicomponent hierarchical porous hollow spheres were observed by SEM (Carl Zeiss Jena scanning microscope) and TEM (JEOL JEM-2011). Brunner – Emmet – Teller (BET) surface area was measured by  $\text{N}_2$  adsorption at liquid nitrogen temperature using a NOVA4000 automated gas sorption system. X-ray photoelectron spectroscopy (XPS) (ESCALAB 250XI) was used to further verify the chemical composition of the samples.

### Electrochemical measurements

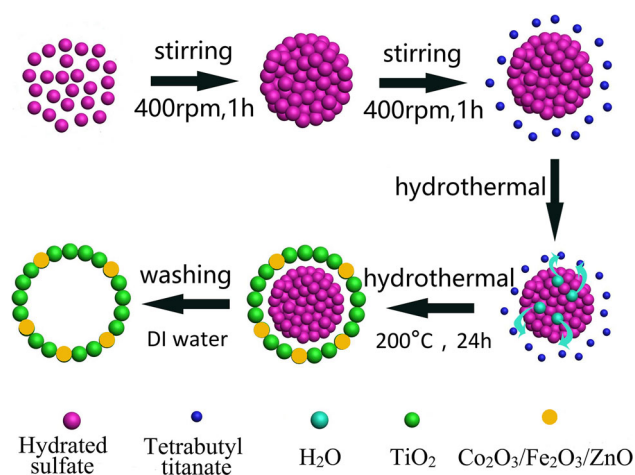
In order to further evaluate the electrochemical performance of the as-prepared  $\text{TiO}_2/\text{Fe}_2\text{O}_3$ ,  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$  hollow spheres as anode materials in lithium-ion batteries, CR2032 coin-type cells were assembled in a glove box (Mikrouna, Shanghai, China) filled with high-purity Ar. Neat lithium metal foil was used as both the counter and reference electrodes, polypropylene microporous film (Celgard 2400) was used as a separator, and a 1-M  $\text{LiPF}_6$  dissolved in a mixture of ethylene carbonate (EC), diethyl carbonate (DEC) and dimethyl carbonate (DMC) (1:1:1 by volume) served as the electrolyte. The working electrodes were fabricated by mixing the active material ( $\text{TiO}_2/\text{Fe}_2\text{O}_3$ ,  $\text{TiO}_2/\text{CO}_2\text{O}_3$  or  $\text{TiO}_2/\text{ZnO}$  hybrid hollow spheres), carbon black and polyvinylidene fluoride (PVDF) in a weight ratio of 75: 15: 10. The slurry was uniformly spread on Cu foil and dried at 110 °C for 12 h in vacuum. The cells were cycled between 0.001 and 2.5 V (vs.  $\text{Li}/\text{Li}^+$ ) at current densities from 0.1C to 5C at room temperature. Cyclic voltammetry (CV) and electrochemical impedance spectroscopy (EIS) were carried out on an electrochemical workstation (CHI660C, Shanghai Chenhua). The CV was performed at a scan rate of 0.1 mV/s. EIS measurements were recorded over a frequency range of 100 kHz to 0.01 Hz.

### Results and discussion

The overall synthesis strategy is illustrated in Scheme 1. First, hydrated sulfate templates were formed by dispersing hydrated sulfates in anhydrous ethanol solution with appropriate stirring speed at which time the hydrated sulfate particles aggregate to form spheres. With the slow addition of tetrabutyl titanate, uniform solutions are obtained. The hydrolysis of tetrabutyl titanate to form  $\text{TiO}_2$  nanocrystallites is triggered by the slow release of water from the hydrated sulfates during hydrothermal treatment. Hollow  $\text{TiO}_2$  spheres were obtained by removing the templates during the washing stage.

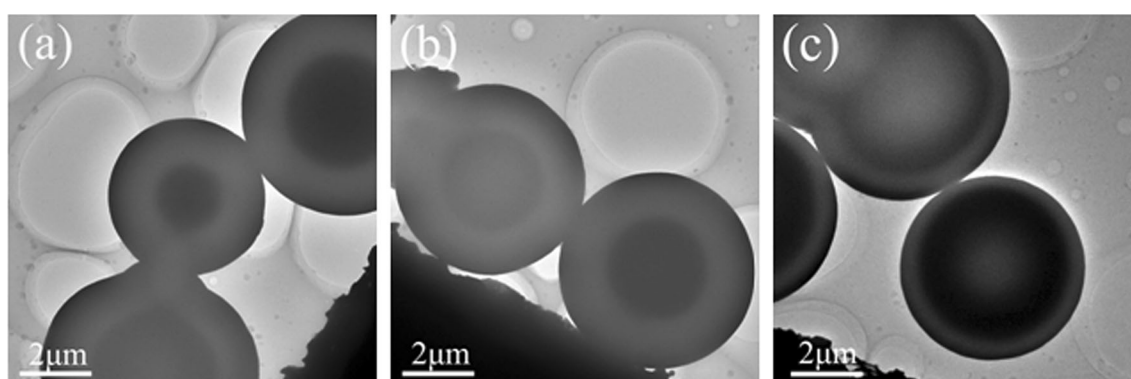
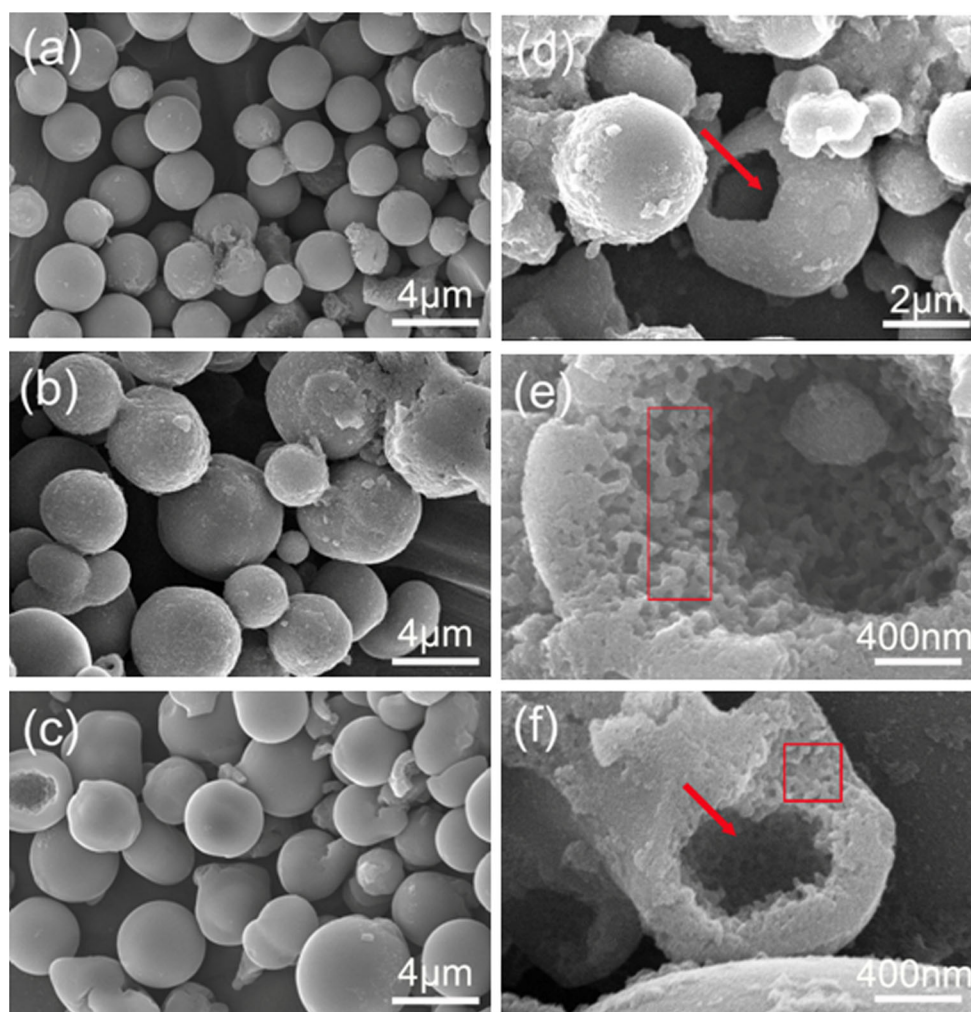
Figure 1 shows SEM images of  $\text{TiO}_2$  hybrid hollow spheres prepared using different templates. The bicomponent  $\text{TiO}_2$  hollow spheres with diameters in the range of 2–4  $\mu\text{m}$  exhibit well-developed spherical morphologies (Fig. 1a–c). SEM images of the  $\text{TiO}_2$  spheres clearly show their hollow structure (Fig. 1d–f). The hollow structure is further supported through TEM imaging (Fig. 2) in which a clear contrast between the edge and the center of the  $\text{TiO}_2$  hollow spheres can be observed. Figure 1e, f shows that the hollow spheres possess a rough internal surface and loosely packed shell which confer high specific surface area. These features are also indicative of the spherical hydrated sulfate templates used in the  $\text{TiO}_2$  sphere formation while also providing additional evidence for the proposed growth mechanism.

The XRD patterns of the as-prepared hybrid hollow spheres are shown in Fig. 3. It can be seen that regardless of the hydrated sulfate used as a template,



**Scheme 1** The synthesis strategy of the  $\text{TiO}_2$  hybrid hollow spheres.

**Figure 1** SEM images of bicomponent hollow spheres prepared by hydrated sulfate templating **a, d**  $\text{TiO}_2/\text{Fe}_2\text{O}_3$ , **b**, **e**  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and **c, f**  $\text{TiO}_2/\text{ZnO}$ .



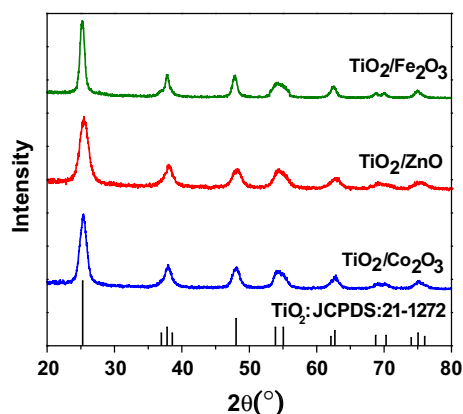
**Figure 2** TEM images of the bicomponent hollow spheres **a**  $\text{TiO}_2/\text{Fe}_2\text{O}_3$ , **b**  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and **c**  $\text{TiO}_2/\text{ZnO}$ .

the diffraction peaks of the as-prepared materials could all be indexed to the anatase phase of  $\text{TiO}_2$  (JCPDS Card No. 21-1272). Furthermore, no evidence of the template materials is observed, indicating successful removal of the template materials. The

average  $\text{TiO}_2$  crystallite size was calculated as 8.5 nm using the Debye–Scherrer equation [39].

The BET-specific surface area of the hierarchical porous  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  hybrid hollow spheres is  $225.25 \pm 3.25 \text{ m}^2/\text{g}$ . The inset in Fig. 4 shows the



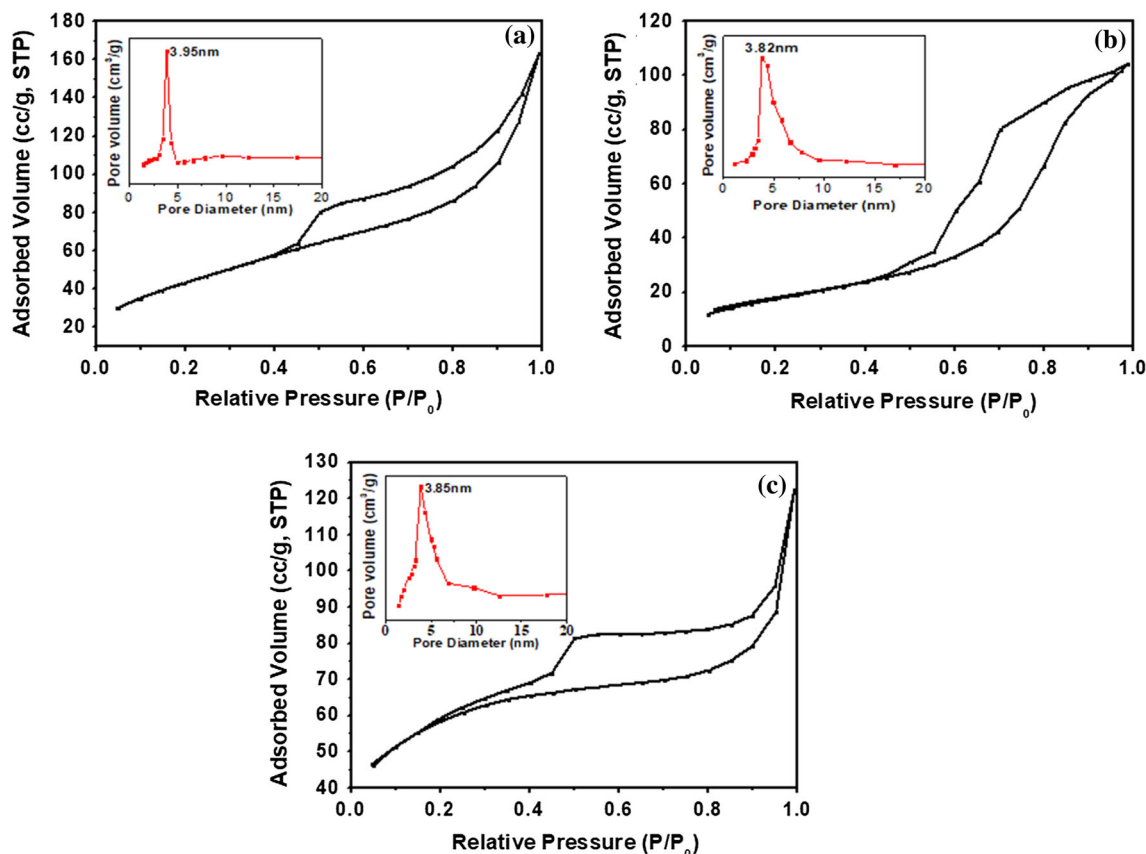


**Figure 3** XRD patterns of the as-prepared hybrid hollow spheres templated by different hydrated sulfates. The JCPDS 21-1272 is corresponding to the anatase  $\text{TiO}_2$ .

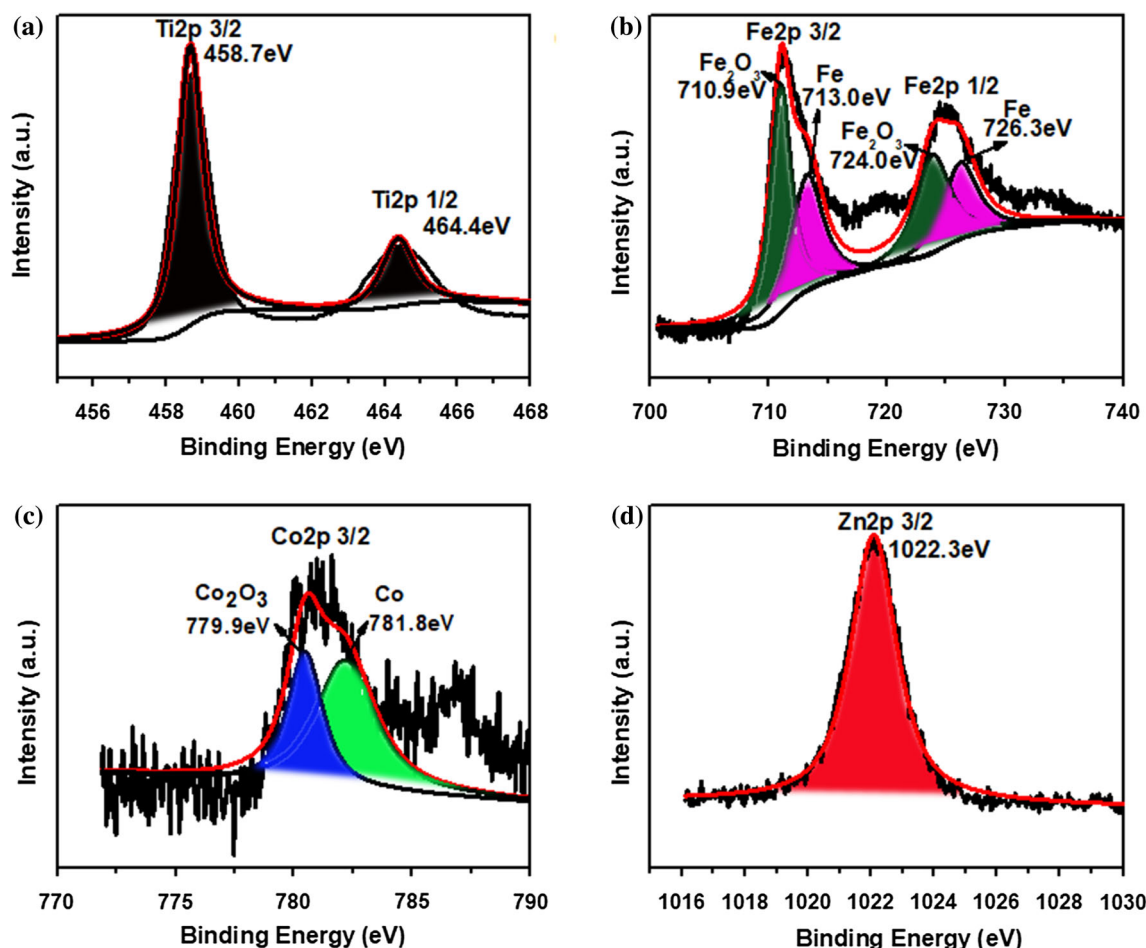
pore size distribution for the hollow  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  spheres, and the Barrett–Joyner–Halenda (BJH) mesopore size distribution of the sample exhibits a sharp peak centered at 3.95 nm. This indicates a relatively uniform mesopore size throughout the hollow

$\text{TiO}_2$  shell. For hollow  $\text{TiO}_2/\text{Co}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$  spheres, the BET measurements also show narrow pore size distributions with average sizes of 3.82 nm and 3.85 nm and large specific surface areas of  $220.69 \pm 2.67 \text{ m}^2/\text{g}$  and  $223.54 \pm 3.58 \text{ m}^2/\text{g}$ , respectively. The consistency of the observed pore sizes and high surface areas across several different metal sulfate hydrates suggests that a hydrated sulfate-assisted hydrothermal method is an ideal templating approach. Both the high specific surface area and hierarchical porous structure of the hybrid spheres are likely contributors to the improved electrochemical performance of lithium-ion battery anode materials during the charge/discharge process as they provide large contact areas for the liquid electrolyte and active materials while also shortening the ion diffusion distance.

In order to further determine the chemical environment of Zn, Fe and Co in the bicomponent materials, XPS was performed (Fig. 5). It can be seen that the Ti 2p XPS spectrum can be fitted with a single



**Figure 4**  $\text{N}_2$  adsorption–desorption isotherms and pore size distributions (inset) of  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  (a),  $\text{TiO}_2/\text{Co}_2\text{O}_3$  (b) and  $\text{TiO}_2/\text{ZnO}$  (c) hollow spheres.

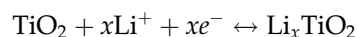


**Figure 5** XPS spectrum of the Ti (a), Fe (b), Co (c) and Zn (d) in the hybrid materials, raw data (black line) and fitted data (red line).

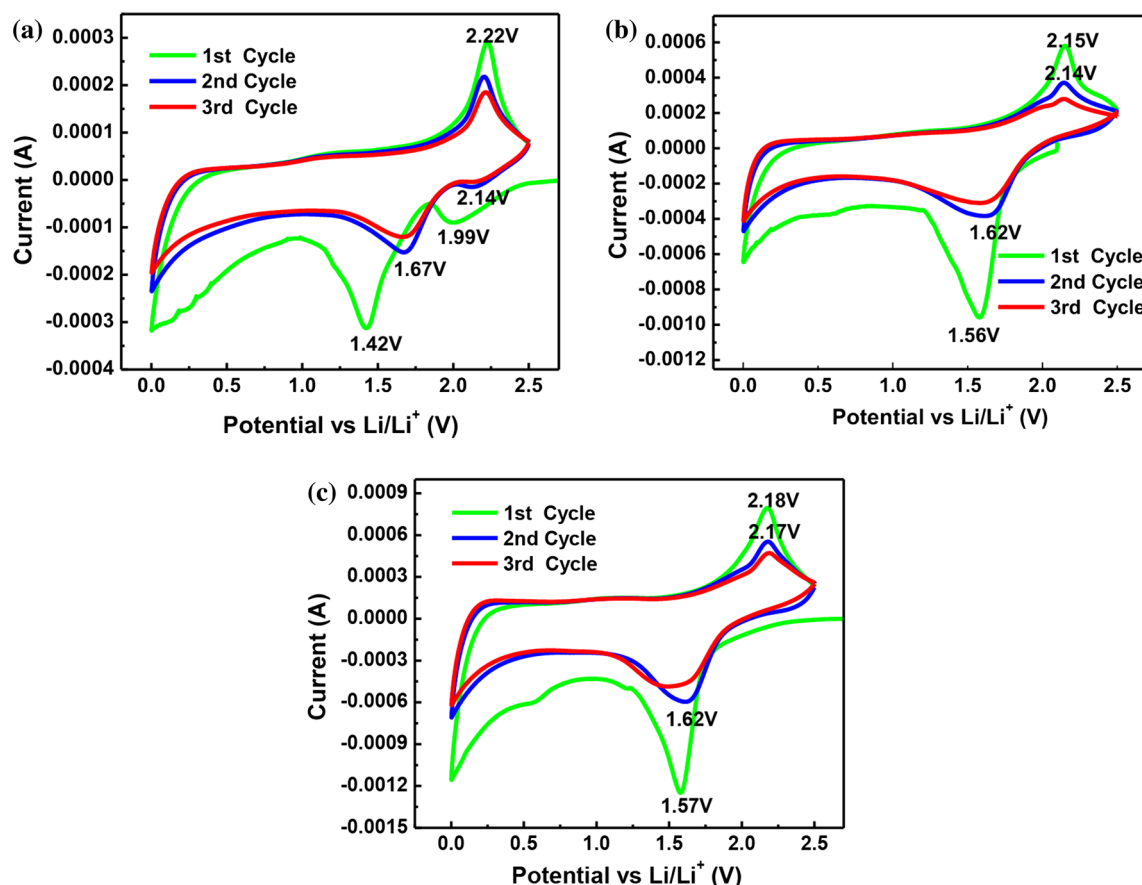
peak corresponding to  $\text{TiO}_2$  (Fig. 5a), and the Zn 2p XPS spectrum can be fitted with a single peak with a binding energy of 1022.3 eV (Fig. 5c) corresponding to ZnO. In contrast, for bicomponent  $\text{TiO}_2/\text{Fe}_2\text{O}_3$ , the Fe 2p 1/2 and 3/2 XPS spectrum can be fitted with two peaks, respectively, and indexed to  $\text{Fe}_2\text{O}_3$  and Fe (Fig. 5b). The Co 2p XPS spectrum can be fitted with two peaks indicating the existence of  $\text{Co}_2\text{O}_3$  and Co in bicomponent  $\text{TiO}_2/\text{CO}_2\text{O}_3$  as expected. XPS results further confirm that the desired bicomponent materials can be obtained by the above one-step hydrothermal method.

Coin cells were assembled to evaluate the performance of the bicomponent  $\text{TiO}_2$  hollow spheres as anodes in lithium-ion batteries. The first three CV curves of the hybrid spheres are presented in Fig. 6. In the first cathodic sweep, the lithium-ion insertion potentials are 1.42 V, 1.56 V and 1.57 V vs.  $\text{Li}/\text{Li}^+$  for  $\text{TiO}_2/\text{Fe}_2\text{O}_3$ ,  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$  bicomponent hollow spheres, respectively. This corresponds

to the reduction of  $\text{Ti}^{4+}$  to  $\text{Ti}^{3+}$  [39]. The subsequent oxidation process occurs at 2.22 V, 2.16 V and 2.18 V vs.  $\text{Li}/\text{Li}^+$  for  $\text{TiO}_2/\text{Fe}_2\text{O}_3$ ,  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$  bicomponent hollow spheres. After the first cycle, the cathodic peak potentials of the three materials all shift to the higher voltage region. In contrast, the anodic peak shifts to the lower voltage region. This suggests a reduced anode polarization and increased  $\text{Li}^+$  insert/extraction reversibility during electrochemical cycling. Concurrently, the corresponding current peaks decrease revealing the existence of an irreversible capacity loss in the initial lithiation–delithiation process. The insertion and extraction of lithium ions into and from the  $\text{TiO}_2$  lattice can be described as follows:



It can also be seen that unlike the CV curves of  $\text{TiO}_2/\text{CO}_2\text{O}_3$  (Fig. 6a) and  $\text{TiO}_2/\text{ZnO}$  hybrid hollow spheres (Fig. 6c), there is a distinct cathodic peak for

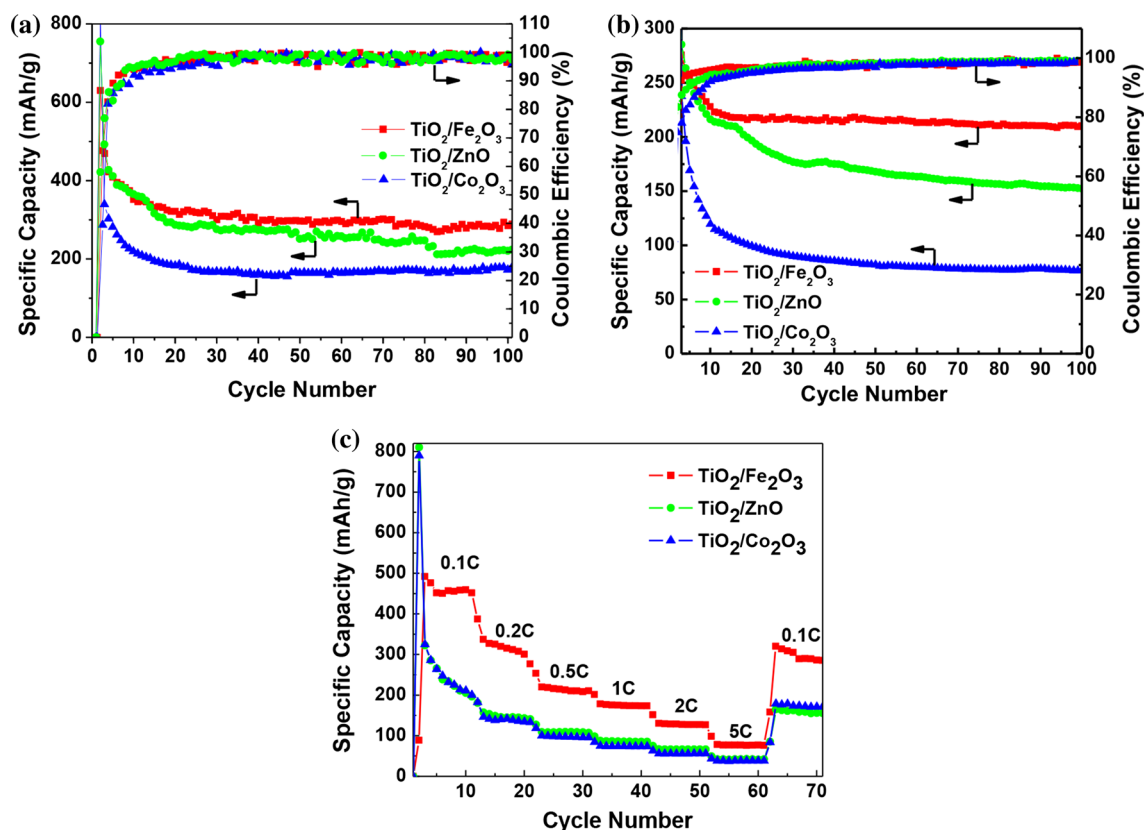


**Figure 6** Cyclic voltammograms for the first (green), second (blue) and third (red) cycles for the  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  (a),  $\text{TiO}_2/\text{CO}_2\text{O}_3$  (b) and  $\text{TiO}_2/\text{ZnO}$  (c) component spheres over the potential range of 0 V to 2.5 V vs.  $\text{Li/Li}^+$  with a scan rate of  $0.1 \text{ mV s}^{-1}$ .

the  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  bicomponent hollow spheres (Fig. 6b). Located initially at 1.99 V in the first cycle, the peak moved to the higher voltage region (2.14 V) in the second and third cycle. This observation may be attributed to lithium insertion into the crystal structure of  $\text{Fe}_2\text{O}_3$  without a significant change in the structure [40]. This would enable an improved lithium-ion storage capability of the bimetallic hollow spheres by essentially providing more sites for lithium uptake. It is noteworthy that the second and third cycles of the voltage–current curves for all three samples almost overlap with each other. This suggests good reversibility and stability of the electrochemical reaction after the first cycle.

Figure 7 presents the cycling and rate performance of the bicomponent hollow spheres. The  $\text{TiO}_2/\text{Fe}_2\text{O}_3$ ,  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$  hollow spheres deliver discharge capacities of 288.6, 180.3 and 221.1 mAh/g, respectively, after 100 cycles at a current density of 0.1C. At a current density of 1C, the hybrid hollow spheres also exhibit stable cycling performance. It can

be seen that the discharge capacity of the three electrodes slightly decreases in the first 10 cycles and then gradually stabilized during the following 90 cycles with a Coulombic efficiency of 99%. After 100 charge–discharge cycles, the  $\text{TiO}_2/\text{Fe}_2\text{O}_3$ ,  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$  bicomponent hollow spheres still deliver discharge capacity of 210.5, 78.6 and 177.7 mAh/g, respectively. These sustained performance values suggest that bicomponent hollow spheres are promising anode materials even when subjected to large current densities (Fig. 7b). Figure 7c presents the rate capability of the bicomponent hollow spheres at various current rates from 0.1C to 5C (1C = 335 mAh/g). The capacity values are the averaged over 10 cycles. As expected, the capacity gradually decreases as the current rate increases. It can be seen that among the three bicomponent hollow spheres, the  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  hollow spheres show the best rate performance with the rate performance of the  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$  hollow spheres being comparable. For  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  hollow spheres,



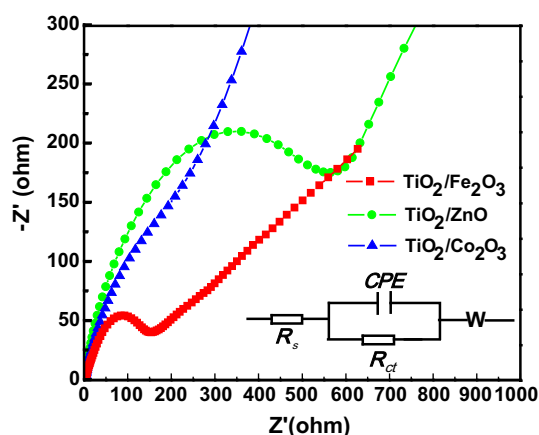
**Figure 7** Cycling performance at a current density of 0.1C (a) and 1C (b), rate performance of bicomponent hollow materials (c):  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  (red),  $\text{TiO}_2/\text{Co}_2\text{O}_3$  (blue) and  $\text{TiO}_2/\text{ZnO}$  (green).

the reversible capacity of the electrode after 50 cycles is 461.1, 307.8, 212.8, 174.3, 127.7 and 79.3 mAh/g for current densities of 0.1C, 0.2C, 0.5C, 1C, 2C and 5C, respectively. More importantly, when the current density returns to 0.1C, a capacity of 288.6 mAh/g can be recovered. This confirms the good rate performance and stability of the bicomponent hollow spheres. Despite  $\text{TiO}_2/\text{Co}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$  hollow spheres possessing lower overall performance values across all current densities, a relatively high discharge capacity can still be recovered by reducing the current density to 0.1C. This observation suggests that the additional metal oxides play the same role in these systems. It is noted that the initial capacity of the three types of hybrid hollow spheres is more than 600 mAh/g, much higher than that of  $\text{TiO}_2$ . The extra capacity may be due to synergistic effect between  $\text{TiO}_2$  and three different metal oxides. Firstly, as we all known that the theoretical capacity of  $\text{Fe}_2\text{O}_3$ ,  $\text{Co}_2\text{O}_3$  and  $\text{ZnO}$  is much higher than that of  $\text{TiO}_2$ , during cycling, the metal oxide will be reduced to metal first, and then, metal will react with lithium to

form alloy, which will contribute more capacity. Secondly, according to Ref. [41], the extra capacity may be due to the further lithium storage via interfacial reaction between the metal and  $\text{Li}_2\text{O}$  phase boundary. And thirdly, the reduced metal may improve the electrical conductivity of the anode, which will further improve the rate performance of the anode.

In order to further understand the enhanced electrochemical performance of the bicomponent hollow spheres, electrochemical impedance spectroscopy (EIS) was used to evaluate the impedance and lithium-ion diffusion rate. As shown in Fig. 8, the EIS spectra are presented in the typical Nyquist plots format showing a semicircle at the high-medium frequency region which can be attributed to the ohmic resistance ( $R_s$ ) and the charge-transfer resistance at the interface between the electrode and electrolyte ( $R_c$ ). The sloping line in the low frequency region corresponds to lithium-ion diffusion in the solid electrode [42]. The Nyquist plots are fitted by using equivalent circuit modeling [43]. In the





**Fig. 8** Nyquist plots of the hollow bicomponent spheres. Inset shows the equivalent circuit used to fit the experimental impedance spectra data.

equivalent circuit (inset),  $R_s$  and  $R_{ct}$  are the ohmic resistance and the charge-transfer resistance of the electrodes. The constant phase-angle element (CPE) and the Warburg impedance ( $W$ ) indicate the lithium-ion diffusion into the bulk active material. The diameters of the resistance arcs for the  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  electrode are much smaller than those for  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$  electrodes. This indicates that the  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  electrode exhibits a lower surface film resistance and charge-transfer resistance. According to the fitting results (Table 1), the  $R_s$ ,  $R_{ct}$  and  $W$  of the  $\text{TiO}_2/\text{Fe}_2\text{O}_3$  bicomponent hollow spheres are smaller than those of  $\text{TiO}_2/\text{CO}_2\text{O}_3$  and  $\text{TiO}_2/\text{ZnO}$ . These results suggest that the electrical conductivity, charge-transfer kinetics and diffusion of lithium ions can be effectively improved by combining  $\text{TiO}_2$  and  $\text{Fe}_2\text{O}_3$ .

## Conclusions

Bicomponent  $\text{TiO}_2$  hollow spheres have been successfully synthesized under hydrothermal conditions using different hydrated sulfates as template. The resulting structures are composed of both  $\text{TiO}_2$  and

the corresponding metal oxide derived from the hydrated sulfate template. The bicomponent hollow spheres are well developed and distinct with a diameter of 2–4  $\mu\text{m}$ . The rough interior surface and loose grain-packed shells afford the hollow spheres large specific surface areas for facilitating lithium-ion diffusion during charging and discharging. These bicomponent hollow spheres can greatly improve the electrochemical properties of  $\text{TiO}_2$ . The composition of the hybrid spheres also has a significant impact on the electrochemical performance of the electrode. The attractive properties and one-step synthesis method make the  $\text{TiO}_2$  bicomponent hollow spheres a promising anode material for lithium-ion batteries.

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## Author contributions

R.P.L and Z.Q.L managed the projects and guided the research. C.S and C.Z conducted all experiments. J.I, Q.W, S.Q.Z and K.J.Y performed the data analysis. R.P.L and Z.Q.L conceived the storyline of the paper. All authors contributed to the discussions and wrote the paper.

## Compliance with ethical standards

**Conflict of interest** The authors declare that they have no conflict interest.

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